



COMPOSITES SCIENCE AND TECHNOLOGY

Composites Science and Technology 63 (2003) 1349–1358

www.elsevier.com/locate/compscitech

# Self-consistent properties of carbon nanotubes and hexagonal arrays as composite reinforcements

R. Byron Pipes<sup>a,\*</sup>, S.J.V. Frankland<sup>b</sup>, Pascal Hubert<sup>c</sup>, Erik Saether<sup>d</sup>

<sup>a</sup>The University of Akron, Polymer Engineering Academic Center, Akron, OH 44325-0301, USA

<sup>b</sup>ICASE, NASA Langley Research Center, Hampton, VA 23681, USA

<sup>c</sup>McGill University, Montreal, Quebec, Canada, H3A 2K6

<sup>d</sup>NASA Langley Research Center, Hampton, VA 23681, USA

Received 4 October 2002; accepted 7 January 2003

#### **Abstract**

A self-consistent set of relationships is developed for the physical properties of single walled carbon nanotubes (SWCN) and their hexagonal arrays as a function of the chiral vector integer pair, (*n*,*m*). Properties include effective radius, density, principal Young's modulus, and specific Young's modulus. Relationships between weight fraction and volume fraction of SWCN and their arrays are developed for the full range of polymeric mixtures. Examples are presented for various values of polymer density and for multiple SWCN diameters.

© 2003 Elsevier Science Ltd. All rights reserved.

Keywords: Single wall carbon nanotubes

#### 1. Introduction

The use of single walled carbon nanotubes (SWCN) for reinforcement of polymeric materials has been given as one of the primary applications for this new material form since its discovery more than a decade ago [1]. Yet no unified basis for reporting the physical properties of SWCN or arrays of SWCN as reinforcements has emerged. For example, the reference volume, required to measure volume dependent properties, is taken as the entire cylindrical volume enclosed by the carbon atom crystal lattice when reporting density [2], while for the effective Young's modulus, the cylindrical volume with inner and outer diameter defined by the bounds of the monoatomic layer of carbon atoms is used [3,4]. These different geometry definitions have led to reported axial moduli ranging from 1.25 TPa [4] for SWCN to 67 GPa [7] in arrays. The need for a self-consistent set of properties is evident. Another author [5] has also stated this need, but does not offer a set of relationships that unify these data. In order to account for the contribution of the SWCN to the overall properties of the polymeric composite wherein the SWCN serves the role of the reinforcement, it is necessary to view its effective properties so that they correspond to a specified volume within the composite. In this work, the representative volume element for all physical properties is defined as that of a cylindrical volume with diameter equal to twice the effective SWCN radius, where the effective SWCN radius includes one-half the van der Waals equilibrium separation distance.

The van der Waals equilibrium distance for SWCN can be defined as the average center-to-center distance between the carbon atom of the SWCN and the nearest atom of the adjacent medium. Therefore, the van der Waals standoff distance is dependent upon the properties of the adjacent medium. In the case of an array composed of SWCN of equal diameter, the individual SWCN are surrounded by other SWCN of like properties and the equilibrium distance is designated as  $\lambda$ . Suspension of SWCN in a polymeric phase yields a different separation distance, v. When employing the equations developed in the present work, it is necessary to establish the value of v corresponding to the suspending medium which can vary widely. In this paper, the equilibrium separation distance in polymers is assumed to be 0.342 nm, the equilibrium separation

<sup>\*</sup> Corresponding author. Tel.: +1-330-972-2702. *E-mail address:* bpipes@uakron.edu (R.B. Pipes).

#### **Nomenclature** vector $a_1$ vector $\mathbf{a_2}$ constant for SWCN principal Young's Amodulus calculation h C-C bond length (nm) constant for SWCN Young's modulus В calculation $E_n$ SWCN Young's modulus (GPa) SWCN Young's modulus in a hexagonal $E_{na}$ array (GPa) SWCN array specific modulus (GPa) $E_{\rho a}$ SWCN specific modulus (GPa) $E_{\rho n}$ Atomic weight $M_{w}$ m number of carbon atoms in a<sub>2</sub> direction number of carbon atoms in a<sub>1</sub> direction nN number of carbon atoms per unit length (atoms/nm) Avogadro's number $N_a$ $R_n$ SWCN radius (nm) $R_{na}$ effective SWCN radius in and hexagonal array (nm) effective SWCN radius (nm) $R_{ne}$ SWCN volume fraction of the hexagonal $V_a$ $V_n$ SWCN volume fraction $W_n$ SWCN array weight fraction SWCN weight fraction $W_n$ Y graphene Young's modulus (GPa) chiral vector function Λ λ distance between the SWCN in an hexagonal array (nm) distance between the SWCN and the ν adjacent surrounding medium (nm) SWCN array density (g/cm<sup>3</sup>) $\rho_a$ SWCN/polymer composite density (g/cm<sup>3</sup>) $\rho_m$ SWCN density (g/cm<sup>3</sup>) $\rho_n$ SWCN density in a hexagonal array (g/ $\rho_{na}$ polymer density (g/cm<sup>3</sup>) $\rho_p$

distance for graphene sheet [14]. The SWCN and SWCN array physical property predictions presented in the present work include density, principal Young's modulus (longitudinal direction) and specific modulus. It is common practice to refer to weight fraction when determining properties of mixtures of SWCN and secondary polymeric phase even though the physical properties of the mixture depend upon the volume fractions of the constituents. In the present work, relationships are developed not only to predict the physical properties

SWCN effective array specific gravity

SWCN effective specific gravity

 $\rho_a$ 

 $\bar{\rho}_n$ 

of the SWCN and its arrays, but also to predict relationships that are necessary to convert SWCN weight fraction to volume fraction in mixtures. These property relationships then serve to provide precursor data for computational methods in the prediction of effective properties of SWCN composites.

The literature is filled with reported values for the Young's modulus of carbon nanotubes and their arrays [6–10]. These reported values differ by almost two orders of magnitude. The SWCN density has also been reported [12,13]; however, no clear methodology for density calculation has been published in the literature.

It is the objective of this paper to develop a self-consistent set of properties for the SWCN and its hexagonal arrays and to provide the mixing rules for conversion of weight fraction to volume fraction for mixtures of SWCN and SWCN arrays with polymers. Properties predicted include effective radius, density, principal Young's modulus, and specific Young's modulus.

## 2. SWCN geometry

The SWCN has been described as a single graphene sheet rolled up with varying degrees of twist as described by its chiral vector,  $C_h$  [14]:

$$C_{h} = n\mathbf{a}_{1} + m\mathbf{a}_{2} \tag{1}$$

where  $\mathbf{a_1}$  and  $\mathbf{a_2}$  are vectors in the two-dimensional hexagonal lattice and the chiral vector,  $\mathbf{C_h}$  is also referred to by its indices, n and m. Nanotubes with chiral vectors of (n,n) and (n,0) have no twist and are classified as chiral nanotubes. These two special cases are sometimes denoted "armchair" and "zig zag," respectively, referring to the pattern of the carbon atoms around the nanotube circumference [14]. Geometric representations of SWCN structures of three different chiralities are shown in Fig. 1. These images illustrate the dependence of the SWCN upon the components of the chiral vector, (n,m). The SWCN radius,  $R_n$  as shown in Fig. 2 is given in the following relation as a function of the integer pair and the C–C bond length, b [14]:

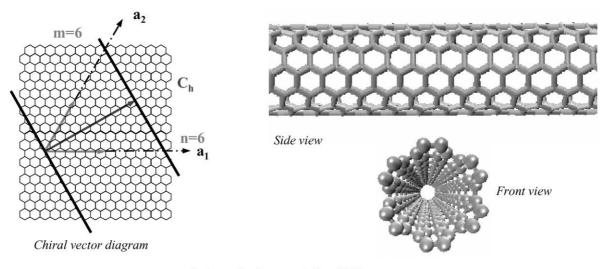
$$R_n = \frac{b}{2\pi} \sqrt{3}\Lambda \tag{2}$$

where

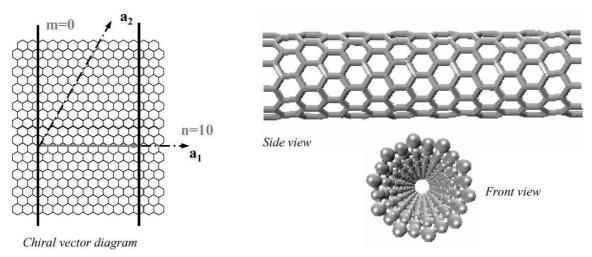
$$\Lambda = \sqrt{(n^2 + m^2 + mn)} \tag{3}$$

For the SWCN, the C–C bond length, *b* is equal to 0.142 nm [14]. The carbon nanotube structure as shown in Fig. 2a is replaced by the effective reinforcing element in Fig. 2b.

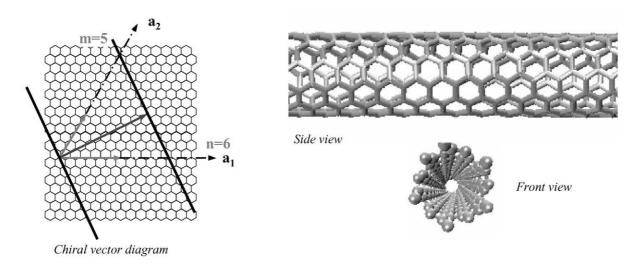
The diameter  $2R_n$  of the SWCN is a strong function of the chiral integers n and m and varies over a wide range as shown in Table 1. However, it is the smaller diameters of SWCN that are of most interest as a polymeric



# a) Armchair nanotube (6,6)

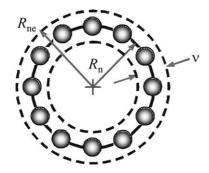


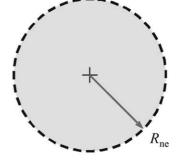
# b) Zig zag nanotube (10,0)



# c) Chiral nanotube (6,5)

Fig. 1. SWCN structure and example of nanotubes.





#### a) Carbon nanotube

# b) Effective reinforcement

Fig. 2. SWCN nomenclature.

reinforcement since the transverse properties of the larger tube diameters can lead to their collapse [15]. The remaining geometric descriptors of the SWCN are the cross-sectional area and length. As stated earlier, several investigators have taken the SWCN equilibrium separation distance to be equal to that of the equilibrium separation distance of graphene sheets of 0.342 nm [15]. This issue requires more discussion since, as stated earlier, the cross-sectional area occupied by the SWCN is influenced by the character of the adjacent medium.

As in discontinuous fiber systems, the length of the SWCN is likely to be a function of the processing technologies used for both the carbon nanotubes and the incorporation of the polymeric phase. Bundle lengths of up to 20,000 nm have been measured [16,17].

The effective radius of the SWCN,  $R_{ne}$ , is defined as the radius given in Eq. (2),  $R_n$ , plus one-half the equilibrium separation distance between the SWCN and the polymer. By defining SWCN radius in this way, the total volume of the heterogeneous mixture is accounted for as follows:

$$R_{ne} = \frac{b}{2\pi} \sqrt{3}\Lambda + \frac{v}{2} \tag{4}$$

Table 1 SWCN diameter for various chiral integers

n	M	$2R_n$ (nm)
5	5	0.68
4	6	0.68
3	7	0.70
2	8	0.72
10	10	1.36
50	50	6.78
6	0	0.48
8	0	0.63
10	0	0.78
12	0	0.94
18	0	1.41
24	0	1.88
50	0	3.91
96	0	7.52

#### 3. SWCN hexagonal array

The synthesis of SWCN typically results in the generation of collimated arrays of SWCN with hexagonal cross-sectional arrangement [17]. In the present study we examine the equilibrium separation distance,  $\lambda$  of such a hexagonal array calculated using the methods discussed in reference [18], wherein the van der Waals interactions of the SWCN are modeled with the Lennard-Jones potential with  $\varepsilon$  = 34.0 K and  $\sigma$  = 0.3406 nm [18]. The results for (n,0) SWCN were calculated in static molecular simulations and are presented in Table 2.

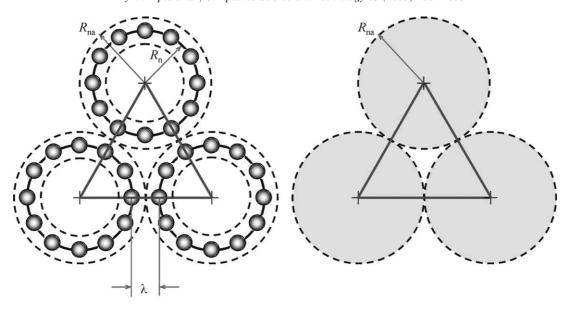
These results suggest that the separation distance is not a function of the SWCN diameter, but rather can be taken as a constant equal to 0.318 nm in agreement with the equilibrium distance of 0.313 nm determined in reference [19] and 0.315 nm in Ref. [20] for a similar range of SWCN diameters. Thus, the SWCN effective radius,  $R_{na}$ , in an array configuration (see Fig. 3) is expressed as:

$$R_{na} = \frac{b}{2\pi} \sqrt{3}\Lambda + \frac{\lambda}{2} \tag{5}$$

The carbon nanotube array shown in Fig. 3a is replaced by the effective reinforcement array shown in Fig. 3b. For a constant separation distance,  $\lambda$  the SWCN volume fraction of the hexagonal array,  $V_a$ , is a

Table 2 SWCN (*n*,0) hexagonal array separation distance

n	Diameter (nm)	Separation distance, $\lambda$ (nm)
6	0.48	0.316
12	0.94	0.317
18	1.41	0.317
24	1.88	0.318
54	4.23	0.318
54 96	7.51	0.319



a) Carbon nanotube array

b) Effective reinforcement array

Fig. 3. SWCN array nomenclature.

constant and is equal to the volume packing fraction of 0.906:

$$V_a = \frac{\pi}{2\sqrt{3}} = 0.906 \tag{6}$$

The hexagonal SWCN array can be considered as a reinforcement form similar to the individual SWCN. As such, the descriptions presented are for the effective properties of the array in the absence of a second phase.

#### 4. Density

The density of the SWCN in the present work is defined as the total mass of the carbon atoms in the enclosed volume as defined in Eq. (4). To calculate the mass of the carbon atoms, it is necessary to express their number per unit length, *N*:

$$N = \frac{4\Lambda}{3b} \tag{7}$$

Then the nanotube density,  $\rho_n$  can be expressed in terms of the chiral vector by combining Eqs. (4) and (7) as follows:

$$\rho_n = \frac{NM_{\rm w}}{\pi N_a R_{ne}^2} = \frac{16\pi M_{\rm w} \Lambda}{3N_{\rm a} b (3b^2 \Lambda^2 + 2\sqrt{3}b\pi \nu \Lambda + \pi^2 \nu^2)}$$
(8)

where  $\nu$ , measured in nm, is the equilibrium standoff distance between the SWCN and the adjacent medium,

 $M_{\rm w}$  is carbon atomic weight and  $N_{\rm a}$  is Avogadro's number. SWCN density as a function of diameter [Eq. (8)] is presented in Fig. 4 along with the specific values of chiral vector. Density is shown in tabular form in Table 3 over a range of chiral vectors. These results show that SWCN density decreases with increasing diameter by an order of magnitude over a range of diameters between 1 and 14 nm.

The results presented in Table 4 for a (10,10) SWCN show that density is a weak function of the choice of the van der Waals distance,  $\nu$ . Nanotubes having other values of chiral vector also showed a similar weak dependence on  $\nu$ .

The density of the hexagonal array of SWCN is the product of the SWCN density in an array configuration

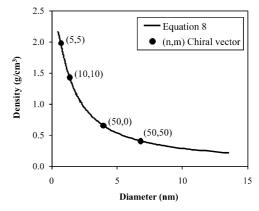


Fig. 4. SWCN density versus diameter.

Table 3 SWCN chiral integers versus density

n	M	Density (g/cm <sup>3</sup> )
5	5	1.99
4	6	1.99
3	7	1.98
2	8	1.95
10	10	1.44
50	50	0.41
6	0	2.18
8	0	2.04
10	0	1.89
12	0	1.75
18	0	1.40
24	0	1.16
50	0	0.66
96	0	0.37

Table 4 van der Waals distance and density for (10,10) SWCN

ν (nm)	Density (g/cm <sup>3</sup> )
0.32	1.471
0.33	1.453
0.34	1.436
0.35	1.419
0.36	1.403

[Eqs. (5) and (7)] and the volume packing fraction of the hexagonal array [Eq. (6)]:

$$\rho_a = V_a \rho_{na} = V_a \frac{N M_w}{\pi N_a R_{na}^2}$$

$$= \frac{\pi}{2\sqrt{3}} \left[ \frac{16\pi M_w \Lambda}{3N_a b (3b^2 \Lambda^2 + 2\sqrt{3}b\pi \lambda \Lambda + \pi^2 \lambda^2)} \right]$$
(9)

Like the density of the individual SWCN, the array density shows a significant reduction with SWCN diameter [Eq. (9)]. It is noteworthy that density of the SWCN array differs from that of the SWCN only by the product of the maximum volume fraction (0.906) and the difference in separation distances,  $\nu$  and  $\lambda$ . Since the separation distances for the array (0.318 nm) and individual SWCN (0.342 nm) are similar, it is likely that the array density will not differ significantly from those of the SWCN. Fig. 5 shows a comparison between density values for individual SWCN and arrays of SWCN for several nanotube chiralities.

## 5. Principal Young's modulus

Here the principal modulus is taken as the modulus of the SWCN and its arrays in the direction parallel to the SWCN longitudinal axis. In the following development

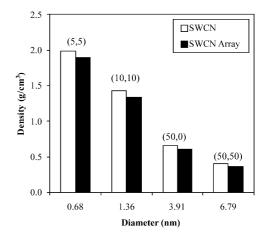


Fig. 5. Comparison of SWCN and array densities.

it is assumed that the SWCN are continuous and that the array consists of SWCN of identical diameter. The present model is a simplified approach wherein the stiffness of a graphene sheet, rolled into the SWCN configuration, is mapped onto the enclosed volume defined by the effective SWCN radius.

#### 5.1. SWCN

The present approach to estimate the Young's modulus,  $E_n$ , of the SWCN is to represent the SWCN as a thin-walled cylinder of outer radius,  $R_{ne}$ , predicted by Eq. (4). It is assumed that the stiffness of the thin-walled cylinder representing the SWCN is equal to that of the Young's modulus of the graphene sheet, Y, and that the SWCN occupies the enclosed cylindrical volume.  $E_n$  and Y are given as:

$$E_n = \frac{8YR_n\nu}{4R_n^2 + 4R_n\nu + \nu^2}$$

$$Y = \frac{\left(C_{11}^2 - C_{12}^2\right)}{C_{11}}$$
(10)

where  $C_{11}$  and  $C_{12}$  are the stiffness constants of graphene [20].

Substituting Eq. (2) into (10) yields:

$$E_n = \frac{4\sqrt{3}\pi b Y \Lambda \nu}{3b^2 \Lambda^2 + 2\sqrt{3}b\pi \Lambda \nu + \pi^2 \nu^2}$$
 (11)

Eq. (11) shows that the modulus of the SWCN decreases with increasing radius. Earlier work described in references [7,10,11] has shown a similar behavior. Here the work described in [10] is compared with the present work predicted in Eq. (11).

$$E_n = \frac{A}{R_n} + B \tag{12}$$

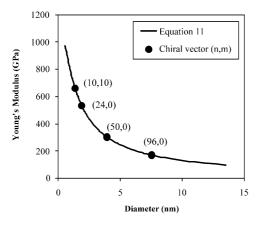


Fig. 6. SWCN Young's modulus versus diameter.

where the constants in Eq. (12) are defined as A = 429.6GPa-nm and B = 8.42 GPa and were obtained from [10]. The choice of the equilibrium separation distance,  $\nu$  and Young's modulus of the graphene sheet in Eq. (11) uniquely determine the quantitative values of the SWCN Young's modulus for all radii. In the present study, we take the experimental value of the Young's modulus of the graphene sheet as reported in Ref. [21] as 1029 GPa and the separation distance as that for the graphene sheet of 0.342 nm. Predictions of SWCN modulus as a function of diameter using Eq. (11) are shown in Fig. 6. Table 5 illustrates the utility of Eq. (11) in determining Young's modulus for five specific sets of chiral integers, (n,m). In addition, a comparison with the results of Refs. [7] and [11] show that all four models provide very similar results.

## 5.2. SWCN hexagonal array extension

Utilizing the Young's modulus,  $E_{na}$ , of individual SWCN in an array configuration, we obtain the following SWCN array modulus:

$$E_a = V_a E_{na} = \frac{2\pi^2 b Y \Lambda \lambda}{3b^2 \Lambda^2 + 2\sqrt{3}b\pi \Lambda \lambda + \pi^2 \lambda^2}$$
 (13)

Results for the principal modulus of the SWCN array shown in Fig. 7 and predicted by Eq. (13) show that the array modulus follows identical trends as for the individual SWCN. This is true because the two predictions

Table 5 SWCN Young's modulus for specific chiral integers

74	100	$E_n$ (GPa)	$E_n$ (GPa)	E <sub>n</sub> (GPa)	$E_n$ (GPa)
n	m	Eq. (11)	$E_n$ (GFa) Eq. (12)	[7]	$E_n$ (GFa) [11]
10	10	662	642	608	680
18	0	647	618	594	664
24	0	536	465	492	550
50	0	304	228	279	311
96	0	171	123	157	175

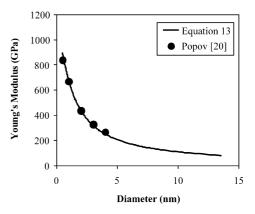


Fig. 7. SWCN array Young's modulus versus diameter.

differ only by the product of the maximum volume fraction (0.906) and the difference in van der Waals distances ( $\lambda$ =0.318 nm for the array). Predictions of Eq. (13) also show excellent agreement with those presented in reference [20] obtained from lattice dynamics calculations for (3n,3n) "armchair" and (3n,0) "zigzag" SWCN as shown in Fig. 7. The data taken from [20] were for four data points and one interpolated value. To further clarify the differences in moduli between the SWCN and the array, results are presented in Fig. 8 for several nanotube chiralities.

#### 6. Specific axial modulus

The specific modulus,  $E_{\rho n}$ , of the SWCN is defined as the ratio of the principal Young's modulus,  $E_n$ , divided by a quantity analogous to specific gravity of the SWCN,  $\bar{\rho}_n$ . Combining Eqs. (8) and (10) yields the expression:

$$E_{\rho n} = \frac{E_n}{\bar{\rho}_n} = \frac{3N_a \sqrt{3}b^2 Y \nu}{4M_{\rm w}}$$
 (14)

Eq. (14) yields results independent of the SWCN diameter since the same SWCN volume is utilized in the calculation of both modulus and density.

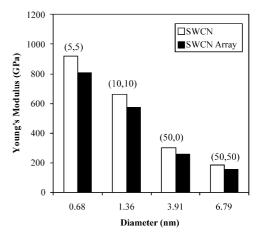


Fig. 8. Comparison of SWCN and array Young's modulus.

In a similar manner, the specific modulus for the SWCN array,  $E_{\rho a}$ , is obtained by combining Eqs. (9) and (13):

$$E_{\rho a} = \frac{E_a}{\bar{\rho}_a} = \frac{3N_a\sqrt{3}b^2Y\lambda}{4M_{\rm w}} \tag{15}$$

These results show that the specific modulus of the SWCN array is identical in form to that of the SWCN.

## 7. Weight fraction versus volume fraction

With the establishment of the density of the SWCN, it is now possible to develop a relationship between weight fraction and volume fraction of SWCN in a mixture with a second material such as a polymer. Consider a SWCN/polymer mixture of density,  $\rho_m$  with a SWCN volume fraction,  $V_n$ , a SWCN density of  $\rho_n$  and a polymer density of  $\rho_p$ . The SWCN volume fraction can be expressed as follows:

$$V_n = \frac{\rho_m - \rho_p}{\rho_n - \rho_p} \tag{16}$$

It is also possible to express the SWCN volume fraction in terms of its weight fraction,  $W_n$ :

$$V_{n} = \frac{\rho_{m}}{\rho_{n}} W_{n} = \frac{W_{n} \rho_{p}}{W_{n} \rho_{p} + (1 - W_{n}) \rho_{n}}$$
(17)

Combining Eqs. (20) and (8) we obtain the final expression for SWCN volume fraction:

$$V_n =$$

$$\frac{W_n 3b (3b^2 \Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\nu^2)\rho_p}{W_n \rho_p 3b (3b^2 \Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\nu^2) + (1 - W_n)16\pi k\Lambda}$$
(18)

The relationship between volume fraction and weight fraction for SWCN-polymer mixtures is shown in Fig. 9 for (6,6), (12,12) and (18,18) SWCN for a polymer density

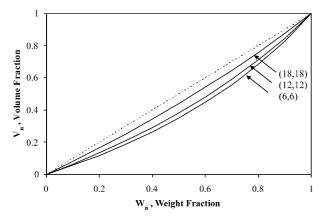


Fig. 9. SWCN volume fraction versus weight fraction (polymer density =  $1~g/cm^3$ ).

of 1 g/cm<sup>3</sup>. These results show that as the diameter of the SWCN is decreased, the non-linearity in the relationship increases. This occurs because the difference between the SWCN and polymer densities increases as the SWCN diameter decreases. Results in the dilute concentration regime ( $W_n < 0.012$ ) are shown in Fig. 10 for convenience since many applications are in this range. The influence of variations in polymer density upon the relationship between volume and weight fraction is shown in Fig. 11 for the (10,10) SWCN. Here increasing polymer density decreases the non-linearity in the relationship. This occurs because as the polymer density increases from 0.8 to 1.2 g/ cm<sup>3</sup>, it approaches the density of the (10,10) SWCN which is 1.44 g/cm<sup>3</sup>. Again, as the density difference between the polymer and SWCN decreases, the non-linearity in the weight fraction volume fraction is also decreasing.

The relationship for the volume fraction of SWCN arrays,  $V_a$ , in terms of the corresponding weight fraction,  $W_a$ , for SWCN arrays in a polymer mixture is:

$$V_a =$$

$$\frac{W_a 3\sqrt{3}b \left(3b^2 \Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\lambda^2\right)\rho_p}{W_a \rho_p 3\sqrt{3}b \left(3b^2\Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\lambda^2\right) + (1 - W_a)8\pi^2k\Lambda}$$
(19)

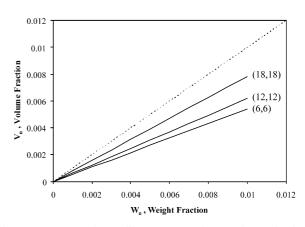


Fig. 10. SWCN results at dilute concentrations (polymer density =  $1 \text{ g/cm}^3$ ).

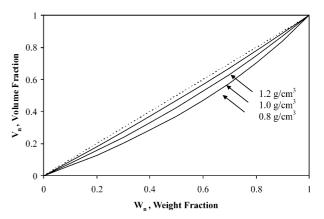


Fig. 11. SWCN (10,10) results for different polymer densities.

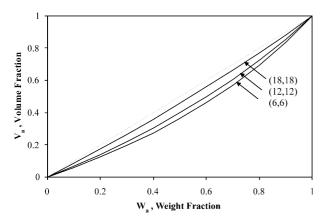


Fig. 12. Array volume fraction versus weight fraction (polymer density  $= 1 \text{ g/cm}^3$ ).

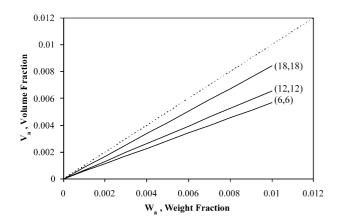


Fig. 13. Array results at dilute concentrations (polymer density =  $1 \text{ g/cm}^3$ ).

Fig. 12 illustrates a similar weight-volume fraction relationship for SWCN arrays as was shown in Fig. 9 for individual SWCN suspended in a polymeric phase of density 1 g/cm<sup>3</sup>. Arrays in dilute concentrations suspended in a polymeric phase are illustrated in Fig. 13 for (6,6), (12,12) and (18,18) SWCN. Finally, the influence of polymer density upon the relationship between polymer mixtures containing SWCN arrays is shown in Fig. 14.

These relationships, based on a consistent definition of volume and thereby, volume fraction, are necessary for micromechanics calculations to determine the effective properties of nanotube reinforced polymers.

#### 8. Summary table of properties

Table 6 shows the set of physical properties of the SWCN and its hexagonal arrays as a function of the chiral vector, (n,m).

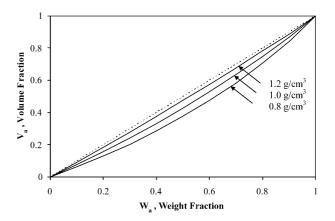


Fig. 14. SWCN (10,10) array results for different polymer densities.

Table 6
Properties of SWCN and SWCN arrays

- Troperties of Swert and Sv	•	
Property	SWCN	SWCN array
Radius (nm)	$R_{ne} = \frac{b}{2\pi} \sqrt{3}\Lambda + \frac{v}{2}$	$R_{na} = \frac{b}{2\pi} \sqrt{3}\Lambda + \frac{\lambda}{2}$
Density (g/cm³)	$\rho_n = \frac{16\pi M_{\rm w}\Lambda}{3bN_a(3b^2\Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\nu^2)}$	$\rho_a = \frac{\pi}{2\sqrt{3}N_a} \left[ \frac{16\pi M_w \Lambda}{3b \left(3b^2 \Lambda^2 + 2\sqrt{3}b\pi \lambda \Lambda + \pi^2 \lambda^2\right)} \right]$
Modulus (GPa)	$E_n = \frac{4\sqrt{3}\pi b  Y \Lambda v}{3b^2 \Lambda^2 + 2\sqrt{3}b\pi \Lambda v + \pi^2 v^2}$	$E_{na} = V_a E_{na} = \frac{2\pi^2 b Y \Lambda \lambda}{3b^2 \Lambda^2 + 2\sqrt{3}b\pi \Lambda \lambda + \pi^2 \lambda^2}$
Specific modulus (GPa)	$E_{\rho n} = \frac{3\sqrt{3}b^2 Y \nu}{4k}$	$E_{\rho a} = \frac{3\sqrt{3}b^2 Y\lambda}{4k}$
Volume-weight fraction	$V_n = \frac{W_n 3b (3b^2 \Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\nu^2)\rho_p}{W_n \rho_p 3b (3b^2\Lambda^2 + 2\sqrt{3}b\pi\nu\Lambda + \pi^2\nu^2) + (1 - W_n)16\pi k\Lambda}$	
	$V_{a} = \frac{W_{a}3\sqrt{3}b\big(3b^{2}\Lambda^{2} + 2\sqrt{3}b\pi\nu\Lambda + \pi^{2}\lambda^{2}\big)\rho_{p}}{W_{a}\rho_{p}3\sqrt{3}b\big(3b^{2}\Lambda^{2} + 2\sqrt{3}b\pi\nu\Lambda + \pi^{2}\lambda^{2}\big) + (1 - W_{a})8\pi^{2}k\Lambda}$	
	$\Lambda = \sqrt{(n^2 + m^2 + mn)}$	
	b = 0.142  nm	$M_{\rm w}/N_{\rm a} = 0.01995$
	v = 0.342  nm	$\lambda = 0.318 \text{ nm}$
	Y = 1029  GPa	

#### 9. Conclusions

The primary objective of this paper was to develop a consistent set of predictions for several of the physical properties of SWCN and their arrays to provide the foundation for developing an understanding of SWCN and arrays as reinforcements in polymers.

The density and moduli of the SWCN and the SWCN array are shown to differ only slightly since the van der Waals distances differ only modestly (0.318 nm versus 0.342 nm) and the SWCN array has a packing fraction of 0.906. The equilibrium separation distance for the SWCN array, determined to be 0.318 nm, was seen to be in good agreement with the values of 0.313 and 0.315 nm published earlier in the literature.

When the simple model for the SWCN principal modulus, consisting of a hollow cylinder with properties of the cylinder wall equal to the stiffness of the graphene sheet was employed, the specific modulus of both the SWCN and its SWCN arrays were found to be independent of SWCN diameter.

For both the SWCN and SWCN array, the principal Young's modulus showed significant dependence upon SWCN radius. Predictions of SWCN array principal moduli were also shown to be in excellent agreement with those previously published in the literature obtained from lattice dynamics calculations for (3n,3n) "armchair" and (3n,0) "zigzag" SWCN. Maximum principal modulus was achieved at the smallest SWCN radius, while density decreased significantly for increased SWCN radius.

A useful equation was derived for the relationship between weight fraction and volume fraction for SWCN and SWCN arrays. The relationship requires knowledge only of the components of the chiral vector of the SWCN and the weight fractions of the constituents to determine volume fraction of the SWCN and SWCN arrays in a mixture.

# Acknowledgements

This research was supported by the National Aeronautics and Space Administration under Contract No. NAS1-97046 while S. J. V. Frankland was in residence at ICASE, NASA Langley Research Center and NASA Cooperative Agreement NCC-1-02002.

#### References

- [1] Iijima S. Helical microbundles of graphitie Carbon. Nature 1991; 354:56
- [2] Collins PG, Avouris P. Nanotubes for electronics. Scientific American 2000:62–9.
- [3] Majumdar A. Not without engineering. Mechanical Engineering 2001:46–9.
- [4] Krishnan A, Dujardin E, Ebbesen TW, Yianilos PN, Treacy MMJ. Young's modulus of single walled nanotubes. Physical Review B 1998;58(20):14013–9.
- [5] Salvetat-Delmotte J-P, Rubio A. Mechanical properties of carbon nanotubes: a fiber digest for beginners. Carbon 2001;40:1729–34.
- [6] Lourie O, Wagner HD. Evaluation of Young's modulus of carbon nanotubes by micro-Raman spectroscopy. Journal of Materials Science 1998;13:2418–22.
- [7] Salvetat J-P, Briggs GAD, Bonard J-M, Bacsa RR, Kulik A, Stockli, Burnham NA, et al. Elastic and shear moduli of singlewalled carbon nanotube ropes. Physical Review Letters 1999; 82(5):944–7.
- [8] Sanchez-Portal D, Artacho E, Soler JM, Rubio A, Ordejon P. Ab initio structural, elastic and vibrational properties of carbon nanotubes. Physical Review B 1999;59:12678–88.
- [9] Hernandez E, Goze C, Bernier P, Rubio A. Elastic properties of C and B<sub>x</sub>C<sub>y</sub>N<sub>z</sub> composite nanotubes. Physics Review Letters 1998;80:4502–5.
- [10] Cornwell CF, Wille LT. Elastic properties of single-walled nanotubes in compression. Solid State communications 1997;101(8):555–8.
- [11] Ruoff RS, Lorents DC. Mechanical and thermal properties of carbon nanotubes. Carbon 1995;33(7):925–30.
- [12] Chesnokov SA, Nalimova VA, Rinzler AG, Smalley RE, Fischer JE. Mechanical energy storage in carbon nanotube springs. Physical Review Letters 1999;82:343–6.
- [13] Gao G, Cagin T, Goddard WA. Energetics, structure, mechanical and vibrational properties of single-walled carbon nanotubes. Nanotechnology 1998;9:184–91.
- [14] Dresselhaus MS, Dresselhaus G, Saito R. Physics of carbon nanotubes. Carbon 1995;33(7):883–91.
- [15] Yakobson BI, Brabec CJ, Bernholc J. Nanomechanics of carbon tubes: instabilities beyond linear response. Physical Review Letters 1996;76(14):2511.
- [16] Arepalli S, Nikolaev P, Holmes W. Production and measurements of individual single-wall nanotubes and small ropes of carbon. Applied Physics Letters 2001;78(10).
- [17] Thess A, Lee R, Nikolaev P, Dai H, Petit RJ, Robert J, et al. Crystalline ropes of metallic carbon nanotubes. Science 1996;273:483–7.
- [18] Saether E, Frankland SJV, Pipes RB. Nanostructured composites: effective mechanical property determination of nanotube bundles. In: Proceedings of the 43rd AIAA/ASME/ASCE/AHS/ASC Structures, Structural Dynamics and Materials Conference, Denver, 2002. Paper no. AIAA–2002–1523.
- [19] Girifalco LA, Hodak M, Lee RS. Carbon nanotubes, buckyballs, ropes and a universal graphitic potential. Physical Review B 2000;62(19):13104–10.
- [20] Popov VN, Van Doren VE, Balkanski M. Elastic properties of crystals of carbon nanotubes. Solid State Communications 2000; 114:395–9.
- [21] Popov VN, Van Doren VE. Elastic properties of single-walled carbon nanotubes. Physical Review B 2000;61(4):3078–84.